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# Experimental Evidence for Two-Dimensional Ostwald Ripening in Semiconductor Nanoplatelets

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**ABSTRACT:** Colloidal semiconductor nanoplatelets are rectangular, quasi-two-dimensional crystallites that are only a few atomic layers thick. In the most heavily studied systems (such as CdE with E = S, Se, or Te), the highly anisotropic nanoplatelet shape forms from an isotropic cubic crystal lattice. This has been difficult to reconcile with standard nanocrystal growth models. Previously, we proposed that nanoplatelets arise due to an intrinsic kinetic instability that enhances growth on narrow crystal facets under surface-reaction-limited conditions. Here, we test this model experimentally by synthesizing small “baby” CdS nanoplatelets and performing ripening experiments. During heating, we observe transitions to thicker nanoplatelet populations that are consistent with two-dimensional Ostwald ripening. We then heat CdSe nanoplatelets in the presence of “baby” CdS nanoplatelets to form CdSe–CdS core–crown nanoplatelets. This indicates that individual nanoplatelets dissolve and transfer their material to thicker nanoplatelets, consistent with the kinetic instability model. Finally, we grow thin films of CdS nanoplatelets by combining “baby” CdS nanoplatelets with cadmium carboxylate and heating on a substrate. This shows that Ostwald ripening may also be exploited as a facile and versatile approach to nanoplatelet growth.

## INTRODUCTION

Methods to prepare nanometer-scale semiconductor crystallites via colloidal chemistry are now highly developed.<sup>1,2</sup> In particular, nanocrystals that exhibit a quasi-spherical shape (typically referred to as quantum dots, QDs<sup>3</sup>) have been prepared from many materials. QDs have been investigated because their electronic transitions are strongly dependent on their diameter,<sup>4,6</sup> providing useful size-tunable optical properties (e.g., absorption and fluorescence). However, if the QD size varies within a sample, spectral broadening occurs that is detrimental for many applications. Consequently, much work has aimed at reducing the QD size distribution during growth.<sup>1,2</sup> This includes efforts to understand and reduce the negative effect of Ostwald ripening.<sup>7</sup> Such ripening is caused by the size-dependent solubility of nanocrystals. Under conditions of depleted reactants, smaller particles in a sample can dissolve and shrink while larger particles remain stable and grow. Because this broadens (or “defocuses”) the size distribution, synthetic protocols to avoid Ostwald ripening have been developed.<sup>8</sup> Nevertheless, the size distribution in real QD samples is never fully eliminated. Even in the best cases, the diameter varies by ~5%.

Such residual size inhomogeneities in semiconductor nanocrystals have been assumed to be unavoidable. Indeed, as other nanocrystal shapes have been introduced over the years (nanorods,<sup>9</sup> nanowires,<sup>10–12</sup> tetrapods,<sup>13–15</sup> cubes,<sup>16</sup> etc.), each has suffered from distributions in size. However, a decade ago, an exception was discovered—semiconductor nanoplatelets (NPLs).<sup>17</sup> These nanocrystals are rectangular plates with a uniform atomic-scale thickness. Moreover, samples can be prepared in which all the NPLs have the same thickness (e.g., 3 monolayers). Because their optical properties are governed by this thickness, they can exhibit better optical properties than QDs, including narrower fluorescence

linewidths<sup>18</sup> and larger absorption cross sections.<sup>19</sup> Thus, NPLs have been studied not only to understand the optical<sup>20</sup> and electronic<sup>21</sup> properties of quasi-two-dimensional (quasi-2D) semiconductors, but also for applications in light-emitting devices<sup>22</sup> and lasers.<sup>23</sup>

Since the first report of CdSe NPLs,<sup>17</sup> subsequent research has led to the control of their lateral shape,<sup>24</sup> the development of new NPL materials (e.g., lead and zinc chalcogenides<sup>25</sup> and mercury telluride<sup>26</sup>), and the growth of NPL heterostructures (e.g., core–crown<sup>27</sup> and core–shell<sup>28–30</sup>). During these developments, other quasi-2D semiconductor nanocrystals, including nanoribbons,<sup>31</sup> nanosheets,<sup>32</sup> and quantum belts<sup>33</sup> have also been studied in parallel. However, during all of this work, the formation mechanism behind NPLs and their uniform thickness remained poorly understood. For CdSe NPLs exhibiting the zincblende (cubic) crystal structure, oriented attachment of preformed nanoclusters to a growing crystallite has been hypothesized,<sup>33</sup> but experimental results suggest continuous lateral growth from CdSe “monomers.”<sup>34–36</sup> For this latter growth mode, it is then unclear how an isotropic crystal lattice generates such a highly anisotropic quasi-2D shape. Templated growth within soft molecular lamellae was proposed as a possible route to break symmetry,<sup>37</sup> or selective binding of at least two different ligands to the surface facets of isotropic crystalline seeds.<sup>18</sup> However, quantitative models based on these ideas have not yet appeared.

Another puzzle has been why the thickness distribution in NPL samples does not broaden during growth. Prolonged heating of reaction mixtures simply causes NPLs to increase slowly in thickness from  $m$  to  $m+1$  layers.<sup>38–40</sup> (We use the common convention that an  $m$ -monolayer-thick CdE NPL contains  $m+1$  layers of Cd and  $m$  layers of E = S, Se, or Te.) Thus, the thickness of NPL samples can be well controlled. The lack of broadening also implies that NPLs

somehow avoid the negative impact of Ostwald ripening.

Due to the many open questions, we have recently sought to clarify the NPL growth mechanism. Our efforts were initiated by a simple experiment that provided an important clue.<sup>40</sup> Namely, we found that zincblende CdSe and CdS NPLs would form in simple solvent-free melts that contained only the chalcogen and a single cadmium carboxylate. This indicates (in agreement with other results) that neither a soft template nor selective binding of different ligands is necessary to obtain NPL growth.

After considering many possible mechanisms, we proposed a simple model based on a kinetic instability.<sup>40</sup> We showed that under certain conditions, lateral growth on a narrow (few-monolayer-thick) “side” facet of a NPL should be orders of magnitude faster than growth on the wide “top” or “bottom” facets. This can occur because the kinetic barrier for nucleation of an additional monolayer should be significantly reduced on a narrow facet. To be “narrow” the facet must have a dimension smaller than the critical island size for growth on a bulk surface of the same material. Indeed, this picture could immediately explain why CdE NPLs had been synthesized only up to a certain thickness.<sup>41–43</sup> For NPLs with thicknesses comparable to or larger than the critical island size, lateral growth would have the same growth rate as that on the wide facets (i.e. on bulk surfaces). Isotropic growth would then be expected.

The above model could also produce specific quantitative predictions that were consistent with experiments. First, in the standard reactions containing only cadmium carboxylates and a S, Se, or Te precursor, it could reproduce the specific thickness ranges obtained in syntheses of NPLs of CdS [2 to 5 monolayers (MLs)<sup>41</sup>], CdSe (2 to 5 MLs<sup>41</sup>), and CdTe (2 to 8 MLs<sup>39,41,44</sup>). Second, it could explain why the same reactants could lead to either QDs or NPLs, depending on the conditions. The kinetic instability requires nanocrystal growth to be surface-reaction limited. If diffusion of reactants is limiting, spherical QDs should result. Therefore, a synthesis with the same precursors, e.g. cadmium myristate and elemental selenium, can yield QDs or NPLs depending on the local concentration of the precursors. NPLs grow if the reactants are highly concentrated, which is the case in the solvent-free melt. In the original CdSe NPL synthesis,<sup>17</sup> a short-chain cadmium carboxylate was required to initiate the NPL growth. We have shown that this step causes the Cd precursor to phase separate as droplets from the organic solvent (octadecene).<sup>40</sup> This concentrates the reactants.

To test the intrinsic-stability model further, we then considered NPL growth under conditions in which the reactants are depleted (i.e. the supersaturation of monomers is reduced).<sup>45</sup> In this case, effects due to Ostwald ripening should arise. Because the standard theory of Ostwald ripening assumes particle shapes that are in thermodynamic equilibrium,<sup>7,45,46</sup> this theory cannot be directly applied to the non-equilibrium shapes of NPLs. Thus, we developed a microscopic model that included the kinetics and thermodynamics of attachment and detachment of CdSe monomers to different facets of a NPL as a function of monomer concentration.<sup>45</sup> Simulations revealed an interesting 2D Ostwald ripening process that included several important stages: (i) Early in the simulation, small isotropic nuclei of many different sizes appeared (e.g., cubes containing 2x2x2, 3x3x3, and 4x4x4 monomers). Random attachment of additional monomers to these nuclei was sufficient to break symmetry and drive anisotropic growth via the kinetic instability. (ii) The first NPLs to appear out of this population of nuclei were the 2-ML-thick NPLs. 2-ML nuclei grow laterally the quickest because their narrow facets exhibit the fastest growth rate (compared to all other growth rates, including the narrow facets on

thicker nuclei). (iii) As the monomers are consumed by NPL growth, 2D Ostwald ripening in the 2-ML NPL fraction is observed. 2-ML facets have the highest solubility compared to thicker facets. Moreover, within a population of 2-ML NPLs, the solubility of the narrow facet increases as the NPL becomes smaller laterally. Thus, as the monomer concentration decreases, the laterally small 2-ML NPLs will eventually become unstable and start to shrink, donating their material to laterally larger 2-ML NPLs, which remain stable and expand. This can be thought of as a 2D analog of the conventional Ostwald ripening. (iv) Eventually, the monomer concentration falls sufficiently that the narrow facets on all 2-ML NPLs become unstable, and this thickness population disappears. (v) The material is then transferred to the 3-ML-thick nuclei, which repeat the same sequence described above for the 2-ML NPLs. They expand, exhibit 2D Ostwald ripening, and eventually disappear. (vi) This process repeats sequentially for 4- and 5-ML NPLs.

These six stages describe our basic growth model. An important point is that it predicts that thicker NPLs appear without any growth on the wide facets. Growth (and dissolution) is only occurring on the narrow facets. We also note that the stages can overlap, causing several NPL thickness populations to coexist simultaneously. This is typically observed in real syntheses, in which 2 or more NPL thicknesses can be present at the same time, requiring post-synthetic selective precipitation to obtain a NPL sample of a single thickness. In our simulations a range of thicknesses existed throughout the growth in the form of small nuclei. However, the mass fraction of monomers contained in most of these thickness populations remained small until each cycled through its lateral growth and ripening process. This occurred sequentially from  $m=2$  to 3 to 4 to 5, with some coexistence of  $m$  and  $m+1$  as the mass fraction of the thinner NPL population decreased while that of the next thickness increased.<sup>45</sup> These simulations could then potentially explain how a population of NPLs under prolonged heating can sequentially increase in thickness from  $m$  to  $m+1$  monolayers.

Here, our aim is to test the results of these simulations against careful ripening experiments. Specifically, we examine what happens when laterally small (“baby”) CdS NPLs are heated. We find that such NPLs indeed grow laterally and, at sufficiently high temperatures, transform to thicker NPLs without the addition of any further reactants. This suggests that material must be transferred from smaller NPLs to larger ones. Indeed, we confirm this conclusion by heating laterally large CdSe NPLs in the presence of “baby” CdS NPLs. The result is CdSe–CdS core–crown NPLs, in which CdS has been transferred to the narrow faces of the pre-formed CdSe NPLs. These results support our previously proposed model and give further insight into the microscopic growth mechanism of NPLs.

## EXPERIMENTAL SECTION

**Chemicals.** 1-octadecene (ODE, 90%, #O806), 2-propanol ( $\geq 99.9\%$ , #34965), acetonitrile ( $\geq 99.9\%$ , #34998), cadmium acetate dihydrate [ $\text{Cd}(\text{OAc})_2 \cdot 2\text{H}_2\text{O}$ , 98%, #289159], oleic acid (OA, 90%, #364525), selenium ( $\geq 99.5\%$ , #209651), toluene ( $\geq 99.7\%$ , #89681), trifluoroacetic acid ( $\geq 99.0\%$ , #302031), and trifluoroacetic anhydride ( $\geq 99\%$ , #106232) were purchased from Sigma-Aldrich. Cadmium oxide (99.999%, #48-0800), triethylamine (99%, #157910010), *n*-hexane, absolute ethanol, and acetyl sulfide ( $>95.0\%$ , #A0976) were bought from Strem Chemicals, Acros Organics, Thommen-Furler AG (Rüti b. Bern, Switzerland), alcosuisse (Bern, Switzerland), and Tokyo Chemical Industries, respectively. All chemicals were used as received without

further purification.

**Synthesis of Cadmium Oleate [Cd(oleate)<sub>2</sub>] and Cadmium Myristate [Cd(myristate)<sub>2</sub>].** Cadmium carboxylates were synthesized according to a modified protocol by Hendricks et al.<sup>47</sup> Briefly, 5.75 g of CdO was dispersed in 20 mL of acetonitrile. The mixture was stirred and cooled with a water bath at room temperature, and 0.7 mL of trifluoroacetic acid and 6.2 mL of trifluoroacetic anhydride were added. After 5 min, the water bath was removed and the flask mildly heated until the cadmium oxide had dissolved. In a separate flask, 28.6 mL oleic acid (20.6 g myristic acid), 180 mL of isopropanol, and 14.0 mL of trimethylamine were mixed. The cadmium trifluoroacetate solution was added to the oleic acid (myristic acid) solution. The white precipitate was suction-filtered through a fritted glass funnel and rinsed thoroughly with methanol. The final product was dried in a vacuum oven at 40 °C and stored under ambient conditions.

**Synthesis of “Baby” CdS NPLs.** 90 mL of ODE, 1.3 mL of OA, and 480 mg Cd(OAc)<sub>2</sub>·2H<sub>2</sub>O were mixed in a 250 mL 4-neck round-bottom flask. The mixture was heated to 120 °C under vacuum and then degassed for 2 h. Typically, the pressure in the flask dropped below 0.13 mbar. The flask was then flushed with N<sub>2</sub>, which caused the temperature to rise by a few degrees. Before proceeding, the temperature was re-stabilized at 120 °C. 0.1 mL of acetyl sulfide was then swiftly injected and the reaction mixture heated to 180 °C. When 180 °C was reached, the heating mantle was rapidly removed and the reaction mixture quenched in a water bath. The resulting “baby” NPLs were washed twice by precipitation with ethanol, centrifugation at 8000 rpm (8590 g) for 10 min, and redispersion in *n*-hexane. The final product, which contains laterally small 2- and 3-ML CdS NPLs, was redispersed in 60 mL ODE or 60 mL toluene for the ripening or thin-film experiments, respectively.

**Synthesis of 4-ML CdSe NPLs.** 4-ML-thick CdSe NPLs were synthesized following an adapted protocol from Pedetti et al.<sup>48</sup> Briefly, 24 mg of Se, 340 mg of Cd(myristate)<sub>2</sub>, and 30 mL of ODE were mixed in a 100 mL three-neck flask and degassed for 30 min at room temperature. The flask was then purged with N<sub>2</sub> and the temperature set to 240 °C. When the temperature reached 205 °C, 86 mg of Cd(OAc)<sub>2</sub>·2H<sub>2</sub>O was added to the flask. 10 min after the temperature reached 240 °C, the reaction was rapidly quenched in an ice bath. When the temperature reached room temperature, 2 mL of OA and 30 mL of *n*-hexane were injected to the reaction mixture. After centrifugation at 6000 rpm (4830 g) for 10 min, the supernatant was discarded and the precipitate redispersed in 20 mL *n*-hexane.

**Ripening Experiments.** 8 mL of the “baby” CdS NPL dispersion, 22 mL of ODE, and 35 mg of Cd(OAc)<sub>2</sub>·2H<sub>2</sub>O were mixed in a 50 mL three-neck flask. The mixture was degassed under vacuum at 120 °C for 60 min and then purged with N<sub>2</sub>. The temperature was ramped to the desired temperature (160–190 °C), held for 12 h 50 min, and then cooled to room temperature naturally. The reaction mixture was diluted with 30 mL of *n*-hexane and centrifuged at 3000 rpm (1210 g) for 5 min to remove left-over Cd(OAc)<sub>2</sub>. The NPLs remained in the supernatant and were purified twice by addition of ethanol, centrifugation at 4000 rpm (2150 g), and redispersion in *n*-hexane.

**Synthesis of Thin Films.** 130 mg Cd(oleate)<sub>2</sub> was dissolved in 2 mL of the “baby” CdS NPL dispersion. 150 μL of this dispersion was dropped on a glass slide and spun for 1 min at 800 rpm. The glass slide was then let dry for at least 30 s. This procedure was repeated a total of four times in air. For transmission electron microscopy

(TEM), a SiO<sub>2</sub>-coated silicon nitride grid (Ted Pella, #21531-10) was fixed on a glass slide with Kapton® tape and spun at 3500 rpm. While spinning, 10 μL of the same dispersion was dropped on the grid. This procedure was repeated a total of three times in air. The thin films were then transferred into a N<sub>2</sub>-filled glovebox, put on a pre-heated hot plate (at 180 °C or 200 °C) for 17 h 15 min, and then allowed to cool naturally.

**Transmission Electron Microscopy (TEM).** Samples were prepared by drop casting dilute dispersions in *n*-hexane onto carbon-coated copper grids. TEM micrographs were recorded on a Philips CM12 (operated at 100 kV). Energy dispersive X-ray spectroscopy (EDX) for elemental mapping was performed on an FEI Talos F200X (operated at 200 kV).

**Spectroscopy.** Aliquots were diluted in *n*-hexane and transferred to quartz cuvettes. Absorption spectra were collected on a Varian Cary 50 spectrometer and photoluminescence (PL) spectra on an Edinburgh Instruments FLS 980 fluorometer.

## RESULTS AND DISCUSSION

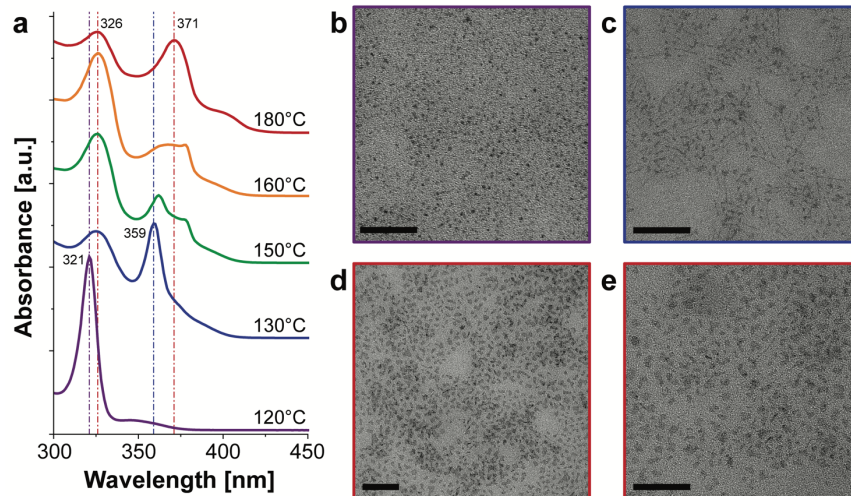
**“Baby” CdS NPLs.** Because we had previously identified acyl chalcogenides as intermediates that form when cadmium chalcogenides are synthesized from a cadmium carboxylate and elemental chalcogen,<sup>49,50</sup> we decided to utilize acetyl sulfide directly as the chalcogenide precursor in our NPL reactions. We modified a previously published NPL protocol<sup>29</sup> by replacing selenium with acetyl sulfide. Following the same temperature profile [ramping to 240 °C with addition of Cd(OAc)<sub>2</sub>·H<sub>2</sub>O at 205 °C] we obtained 4-ML-thick CdS NPLs with a prominent absorption peak at 406 nm.<sup>41</sup> The lateral size of the resulting NPLs was approximately 10 × 20 nm, as determined by TEM (see Section S1 in the Supporting Information). This result just confirms the now expected result that acetyl sulfide can be used to prepare conventional CdS NPLs.

More interesting, the use of acetyl sulfide as the chalcogenide source also allowed us to track the reaction product during the temperature ramp. We exploited this to analyze the growth. Namely, we collected aliquots and measured optical absorption spectra and TEM micrographs on each. Immediately after the injection of acetyl sulfide at 120 °C, a sharp peak at 321 nm appeared (Figure 1a), and small crystallites could be observed in TEM (Figure 1b). This spectral feature, which disappears upon further heating, was previously assigned to a magic-sized cluster (MSC).<sup>51,52</sup> A definitive assignment of this species is challenging due to its small size.

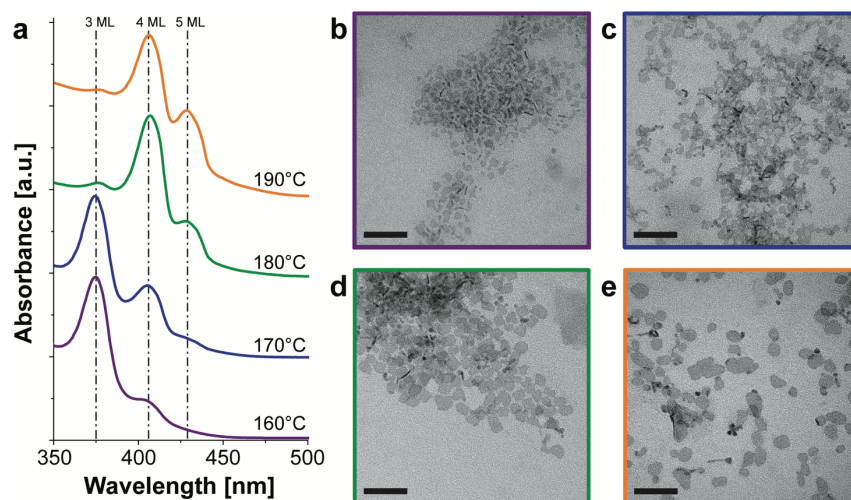
With further heating to 130 °C, two new absorption peaks emerged at 326 and 359 nm. The corresponding TEM micrographs (Figure 1c) show mostly very thin needle-like structures but also some particles that are more isotropic in the lateral dimensions. With increasing temperature to 150 and then 160 °C (Figure 1a), the peak at 326 nm grew, the peak at 359 nm gradually vanished (see Figure S2 in the Supporting Information to support that it is vanishing rather than just shifting), and a new sharp feature at 377 nm appeared. This latter peak has previously been attributed to another CdS MSC.<sup>53</sup> However, it is also possible that the 359 and 377 nm features arise from the needle-like structures. These two absorption peaks disappeared (Figure 1a) together with the needle-like structures (Figure 1d,e) in the 180 °C aliquot.

The remaining two absorption maxima in Figure 1a at 326 and 371 nm appear to arise independently. From this, we conclude that two distinct species are present. However, we could not detect obvious differences between the nanocrystals in the TEM micrographs (Figure 1d,e), other than small and apparently random





**Figure 1.** Synthesis of “baby” CdS NPLs with  $\text{Cd}(\text{oleate})_2(\text{OAc})_{2-x}$  and acetyl sulfide. (a) Absorption spectra of aliquots taken as the temperature is increased after injection of acetyl sulfide. Vertical lines at 321, 326, 359, and 371 nm are provided to compare the spectra. TEM micrographs for aliquots taken: (b) at 120 °C immediately after the injection of acetyl sulfide, (c) at 130 °C, (d) at 180 °C at low magnification, and (e) at 180 °C at higher magnification. Scale bars are 50 nm.



**Figure 2.** Ostwald ripening of “baby” CdS NPLs. (a) Absorption spectra of the CdS NPLs formed as a function of temperature. Corresponding TEM micrographs of the NPLs formed at (b) 160 °C, (c) 170 °C, (d) 180 °C, and (e) 190 °C. Scale bars are 50 nm.

variations in the lateral shape and size. Both absorption maxima occur at wavelengths very close to literature values for the electron/light-hole transitions for 2- and 3-ML CdS NPLs,<sup>41</sup> with a small blue-shift (326 vs. 331 nm and 371 vs. 373 nm, respectively). Blue-shifts in the excitonic transitions of CdSe NPLs have been observed when their lateral sizes were small.<sup>18</sup> Thus, our observations can be explained by the presence of 2- and 3-ML CdS NPLs that are laterally quantum-confined (“baby” CdS NPLs). This conclusion is also consistent with the small nanocrystals that appear in Figure 1d,e.

When the same reaction was conducted at constant temperature, the same evolution was observed. In other words, after degassing we heated the reaction mixture to a certain temperature (140, 160, or 180 °C), and then the injection and growth were performed at this temperature. Higher temperatures only increased the reaction rate (see Section S2 in the Supporting Information).

**Ostwald Ripening.** Based on the results from the previous section, we prepared “baby” CdS NPLs by quenching the reaction at 180 °C and precipitating the product twice with ethanol. To

investigate Ostwald ripening of these NPLs, we then dispersed them in ODE with  $\text{Cd}(\text{OAc})_2 \cdot 2\text{H}_2\text{O}$  and heated them at different constant temperatures. Even though no sulfide precursor was added, we observed lateral growth and transitions to thicker populations, depending on the temperature. In agreement with the model discussed above, the thinnest (2 ML) population was the most unstable, and it rapidly dissolved completely even at relatively low temperatures (<160 °C), as evidenced by the disappearance of the absorption maximum at 326 nm. At temperatures above 160 °C (Figure 2a), we observed a continuous transition from populations with thicknesses of 3 ML (electron/light-hole transition at 374 nm) to 4 ML (407 nm) and then 5 ML (429 nm). While the absorption spectra show different thicknesses of NPLs present at each investigated temperature, the NPLs with different thicknesses could not be distinguished by TEM (Figure 2b-e). The average lateral size of the NPLs increases with temperature, and the lateral size distribution is clearly broad.

When the same ripening experiment was conducted without additional short-chain cadmium carboxylate, i.e. “baby” CdS NPLs

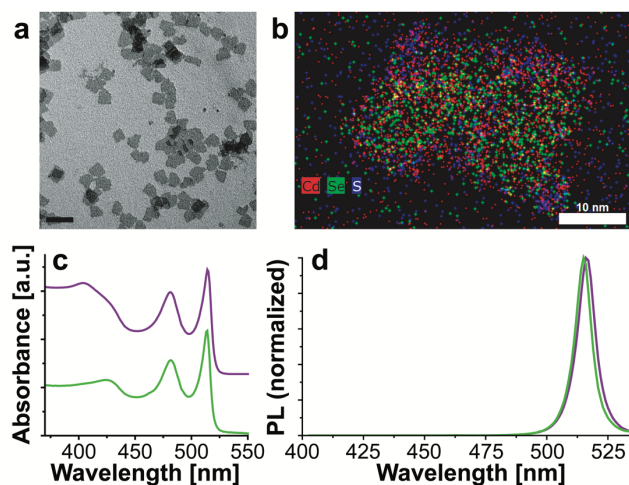
were heated in ODE without  $\text{Cd}(\text{OAc})_2$ , the characteristic absorption peaks (at 374 and 407 nm) disappeared, and featureless absorption grew in at wavelengths longer than 450 nm. This suggests the formation of crystals with random sizes and shapes (see Section S3 in the Supporting Information). This observation is consistent with the prediction that NPL growth requires surface-reaction-limited conditions, which is presumably not maintained here.<sup>46</sup> Without the short-chain  $\text{Cd}(\text{OAc})_2$ , phase-segregated droplets do not form in the ODE.<sup>40,45</sup> Thus, any CdS monomers released by dissolving “baby” NPLs are not concentrated within these droplets, and conditions for 2D growth are not satisfied.

When laterally extended 4-ML CdS NPLs were kept at high temperatures in the presence of  $\text{Cd}(\text{OAc})_2$ , no transitions to thicker populations were observed even after several weeks (see Section S4 in the Supporting Information). This is an important observation for the mechanism behind the transition to thicker NPL populations. It had been hypothesized that the transformation of  $m$  to  $m+1$  MLs for NPLs happens because of surface nucleation and rapid overgrowth of additional layers on the wide “top” and “bottom” facets.<sup>38</sup> The lack of experimentally observed thickness transitions for the 4-ML NPLs is inconsistent with this hypothesis. Moreover, it would be difficult to explain why such a process would stop at  $m+1$  MLs. Rather, growth on the wide facets would presumably lead to a broad range of thicknesses.

By contrast, our kinetic-instability model predicts that growth on the wide facets is highly unlikely compared to lateral growth on the narrow facets. Even if the thermal energy (i.e. the temperature) were increased such that the kinetic barrier for growth on the wide facets could be overcome, this would lead to isotropic growth. Under these conditions, the kinetic instability that drives NPL growth would disappear. Our model also predicts that the dissolution of NPLs should be easier if they are laterally small. Thus, the transformation to thicker populations should be faster for “baby” NPLs, as observed. Laterally extended NPLs can be stable indefinitely depending on the monomer concentration. Therefore, our results agree with the model predictions that transitions between different NPL thicknesses occur via lateral dissolution and growth.

**Nanoheterostructures.** If correct, the growth model above should presumably be valid for any cubic semiconductor that forms NPLs. Therefore, it should be possible to combine materials and form nanoheterostructures such as core–crown NPLs using Ostwald ripening. To test this, we synthesized 4-ML CdSe NPLs with a lateral size of approximately  $10 \times 20$  nm according to a published protocol.<sup>48</sup> We then dispersed these NPLs with  $\text{Cd}(\text{OAc})_2 \cdot 2\text{H}_2\text{O}$  and “baby” CdS NPLs and heated the mixture to  $160^\circ\text{C}$ . The concentrations and temperatures corresponded to the Ostwald ripening experiments for CdS described above, but with the additional presence of CdSe NPLs.

The experiments described in the previous section on 4-ML CdS NPLs suggest that the laterally extended 4-ML CdSe NPLs should be stable (i.e. not dissolving) under the reaction conditions. The “baby” CdS NPLs should also undergo the same 3- to 4-ML thickness transformation as observed without the CdSe NPLs. However, because the CdSe NPLs have the same thickness as the product of the CdS transformation, heterogeneous nucleation of CdS on the narrow facets of the CdSe NPLs could also occur. Thus, we might expect the final product to be a mixture of 4-ML CdS NPLs and 4-ML CdSe–CdS core–crown NPLs. We stress that the goal of these experiments is not to improve synthetic approaches to CdSe–CdS core–crown NPLs, but rather to test our growth model.



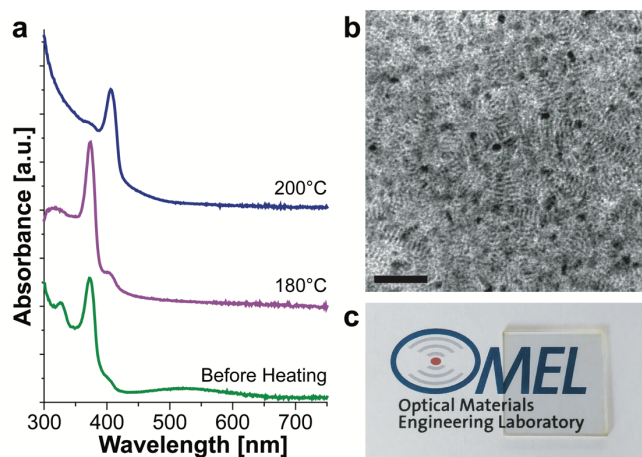
**Figure 3.** Nanostructures formed by heating a mixture of 4-ML CdSe NPLs and “baby” CdS NPLs. (a) TEM micrograph of the NPLs that result after ripening. Scale bar is 40 nm. (b) EDX mapping (red = Cd, green = Se, and blue = S) of two NPLs lying next to each other. Collecting significant signal was challenging due to electron-beam damage. (c) Absorption spectra of the CdSe NPLs before (green) and after (purple) ripening. The product shown after ripening is purified. Further details and data are shown in Section S5 in the Supporting Information. (d) Normalized photoluminescence (PL) spectra of the CdSe NPLs before (green) and after (purple) ripening. Excitation is at 400 nm. A  $\sim 1$  nm peak shift is observed. The above results indicate the formation of CdSe–CdS core–crown NPLs during the ripening process.

Absorption spectra of aliquots taken during the reaction do show the transformation of 3- to 4-ML CdS NPLs. Simultaneously, the electron/light-hole and electron/heavy-hole transitions of the CdSe NPLs remain unchanged (Section S5 in the Supporting Information). By adding small amounts of ethanol to the dispersion, we could selectively precipitate a portion of pure core–crown NPLs. The rest remained in the supernatant along with the 4-ML CdS NPLs. The final, washed precipitate was analyzed by TEM (Figure 3a), EDX (Figure 3b), and optical spectroscopy (Figure 3c,d).

EDX was challenging to collect due to electron-beam sensitivity of the NPLs. This explains the low signal in Figure 3b. Nevertheless, the EDX image, which contains two adjacent NPLs, suggests an accumulation of CdS around the edges of the NPLs. The Se signal appears more in the NPL cores. This suggests that 4-ML CdSe–CdS core–crown NPLs were formed.

This conclusion is also supported by the absorption spectra of the CdSe NPLs before and after heating, which are almost identical except for the appearance of a small local maximum at 405 nm (Figure 3c). This feature, which coincides with the electron/light-hole transition of 4-ML CdS NPLs, also appears in the PL excitation spectrum obtained by monitoring the emission of the CdSe NPLs at 520 nm (Section S5 in the Supporting Information). This indicates that the CdS is co-located with the CdSe NPLs. Finally, the emission intensity maximum of the CdSe NPLs was slightly red-shifted after ripening; no emission from 4-ML CdS NPLs was detected (Figure 3d). All of these data are consistent with prior measurements on CdSe–CdS core–crown NPLs.<sup>54</sup> Thus, these results indicate that the “baby” CdS NPLs dissolved and the material grew on the 4-ML CdSe NPLs. We note that we also attempted to ripen “baby” CdSe NPLs in the presence of CdS NPLs to form CdS–CdSe core–crown NPLs. However, we have not yet found conditions (concentration, temperature, etc.) where the “baby” CdSe NPLs would dissolve





**Figure 4.** Ripening of thin films containing “baby” CdS NPLs and Cd(oleate)<sub>2</sub> deposited on glass substrates. (a) Absorption spectra of films before heating and after heating to either 180 °C or 200 °C. (b) TEM micrograph and (c) photograph of the final film produced. Scale bar is 50 nm. (The OMEL logo, designed by D. Norris, is used with permission.)

while the CdS NPLs were stable.

**Thin Films.** Our growth model also suggests that the same Ostwald ripening will occur in a thin-film matrix that contains only “baby” CdS NPLs and cadmium carboxylate. Such expectations follow directly from our previous work on the formation of NPLs in solvent-free melts.<sup>40</sup> Therefore, we dispersed “baby” CdS NPLs with Cd(oleate)<sub>2</sub> in toluene and spin coated the mixture onto glass slides or TEM grids. After the toluene evaporated a thin solid film was formed. These substrates were then heated above the melting temperature of Cd(oleate)<sub>2</sub>. Because the melt is then essentially a highly concentrated liquid dispersion of “baby” NPLs, CdS monomers released during ripening have easy access to neighboring NPLs. Consequently, we would expect surface-reaction-limited conditions to exist. Consistent with this, when our films were heated to 180 °C and 200 °C absorption spectra (Figure 4a) revealed the same transformation to thicker NPLs that were observed in liquid dispersions (Figure 2).

Presumably due to the high concentration of the resulting NPLs, they also formed stacks with lengths in the range of 100 nm, as observed in TEM (Figure 4b). While more research is necessary to understand the observed stacking, recent publications have shown that CdSe NPLs form large stacks in the presence of oleic acid due to the depletion interaction.<sup>55</sup> A similar mechanism may be responsible here. We also note that thin films prepared with this protocol were transparent and smooth (Figure 4c).

## CONCLUSIONS

We have performed a series of experiments that indicate that “baby” CdS NPLs undergo two-dimensional Ostwald ripening under surface-reaction-limited conditions. These results further support our previously published NPL growth model. In particular, they demonstrate that thickness transformations occur via dissolution of thinner populations. The dissolved material is then re-deposited on thicker NPLs through lateral growth. Our observation of temperature-dependent thickness transitions and of CdSe–CdS core–crown NPLs are inconsistent with both layer overgrowth on the wide facet<sup>18</sup> and oriented attachment.<sup>56</sup> Instead, they support the theoretical prediction that only lateral growth is operable, with the thicker NPLs growing at the expense of thinner ones.<sup>45</sup>

Because NPLs can be grown via Ostwald ripening without being restricted by the kinetics or thermodynamics of side reactions (e.g. reduction of the chalcogen), we believe that future studies on this system with other materials can be useful for providing quantitative data on different NPLs, such as growth temperatures, accessible thicknesses, or whether two-dimensional growth can occur at all. Furthermore, it can potentially enable the synthesis of nanoheterostructures in systems where this is currently not possible, such as the metal halide perovskites.

## ASSOCIATED CONTENT

### Supporting Information

Further synthetic details and experimental data, including absorption, PL, and PL excitation spectra, and TEM micrographs.

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### Notes

The authors declare no competing financial interest.

## ACKNOWLEDGMENTS

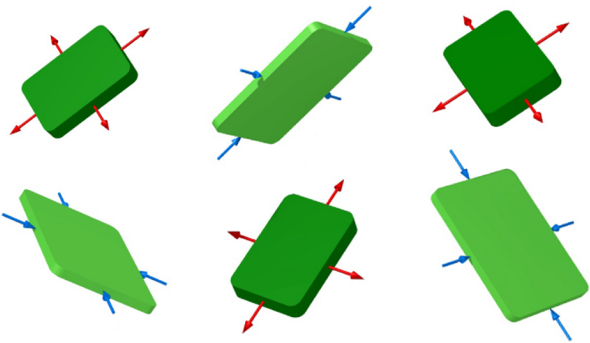
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# Experimental Evidence for Two-Dimensional Ostwald Ripening in Semiconductor Nanoplatelets

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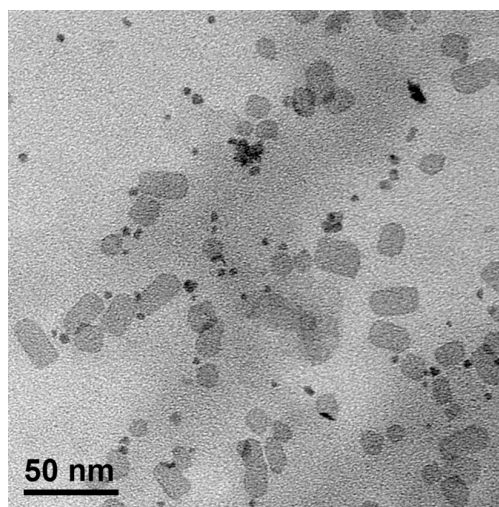
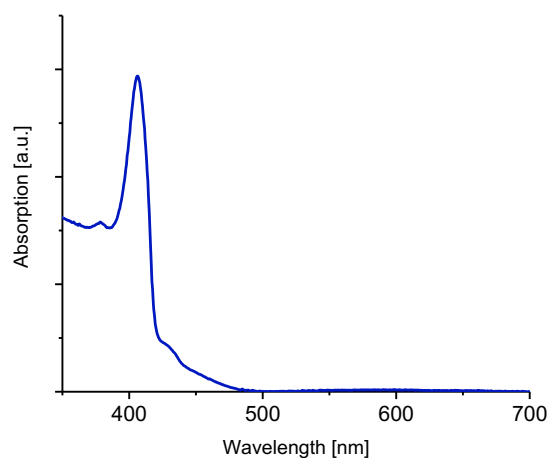
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## Supporting Information

### S1. Synthesis of 4-ML-Thick CdS NPLs with Acetyl Sulfide

We adapted a previously reported synthesis<sup>S1</sup> by replacing the elemental sulfur with an equimolar amount of acetyl sulfide. Briefly,  $\text{Cd}(\text{OAc})_2 \cdot 2\text{H}_2\text{O}$  and OA were dispersed in ODE and degassed for 1 h at 110 °C. The flask was subsequently flushed with  $\text{N}_2$ , acetyl sulfide was injected, and the temperature ramped to 240 °C. When the temperature reached 205 °C,  $\text{Cd}(\text{OAc})_2 \cdot 2\text{H}_2\text{O}$  was added to the reaction mixture. The temperature increase continued to 240 °C, where it was maintained for 15 min. As evidenced by the absorption feature at 406 nm (Figure S1), the synthesis yielded an almost pure population of 4-ML CdS NPLs. TEM revealed NPLs with lateral dimensions of approximately  $10 \times 20$  nm, which is comparable to the results obtained with elemental sulfur as the precursor.

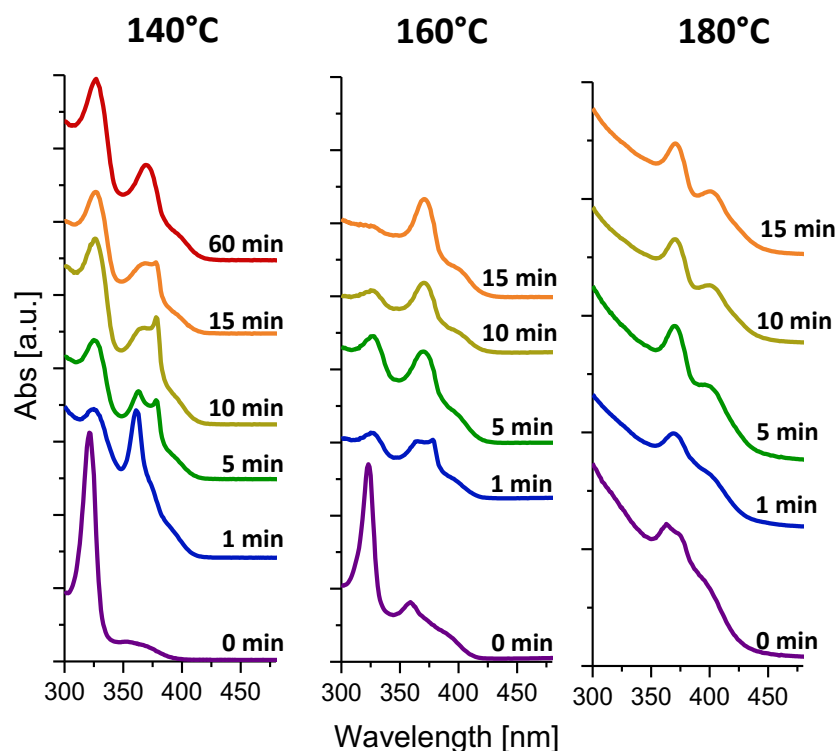


**Figure S1.** Absorption spectrum (left) and TEM micrograph (right) of 4-ML CdS NPLs obtained with acetyl sulfide as the sulfur precursor.

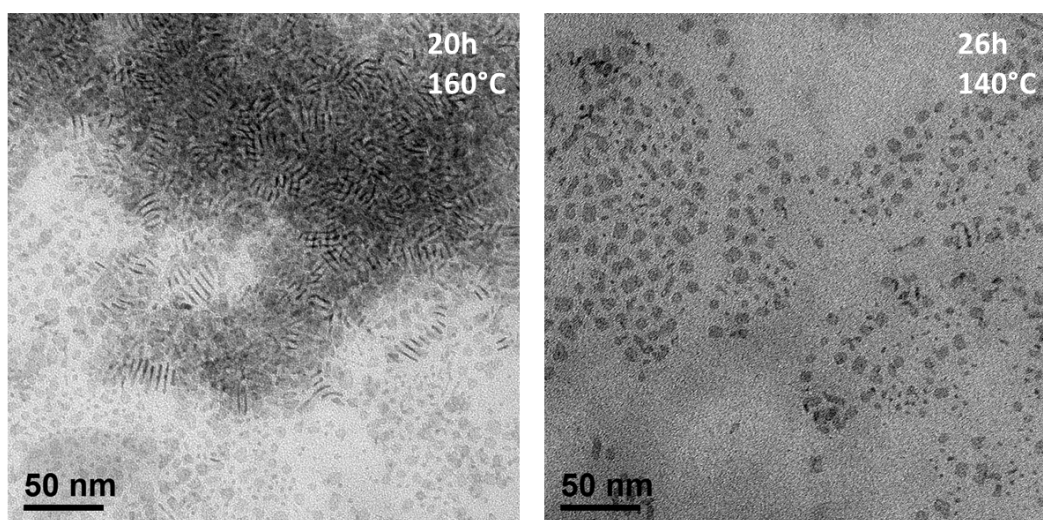


## S2. Isothermal Colloidal Syntheses of “Baby” CdS NPLs

In the main text, we describe the use of acetyl sulfide to track the reaction product formed during the temperature ramp in the synthesis of CdS NPLs. Absorption spectroscopy and TEM microscopy (see Figure 1 in the main text) are used to conclude that 2- and 3-ML CdS NPLs are produced that are laterally small (i.e. “baby” CdS NPLs). Here, the same synthetic protocol was adapted to investigate isothermal conditions. After degassing, the reaction mixture was heated to the desired temperature and kept constant. Acetyl sulfide was swiftly injected. Absorption spectra of aliquots taken at different times and temperatures are shown in Figure S2. At temperatures up to 160 °C, the same reaction stages as described in the main text were observed. The temperature only changed the reaction rate. After sufficiently long times (typically one day), NPLs with lateral extensions of more than 10 nm were observed (Figure S3). At 180 °C, we directly observed the formation of 3- and 4-ML NPLs.



**Figure S2.** Absorption spectra of aliquots taken at different times for isothermal colloidal syntheses of CdS NPLs using acetyl sulfide as the sulfur precursor.

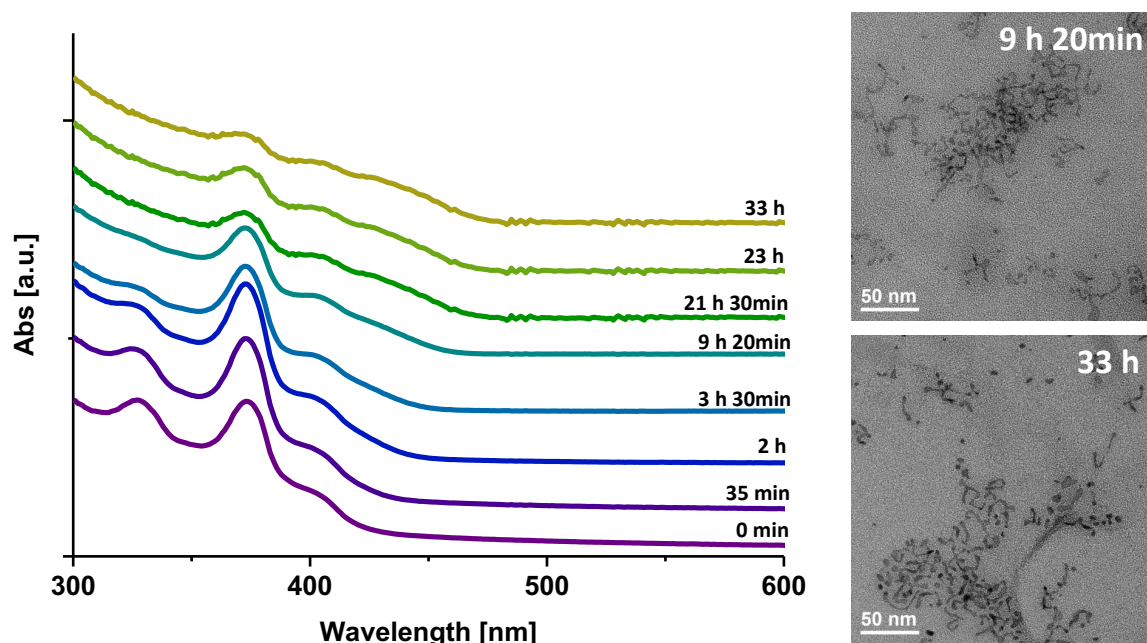


**Figure S3.** TEM micrographs of CdS NPLs synthesized with acetyl sulfide as the sulfur precursor under isothermal conditions. The growth is maintained at 160 °C for 20 h (left) and 140 °C for 26 h (right).



### S3. Heating “Baby” CdS NPLs in ODE without Cd(OAc)<sub>2</sub>

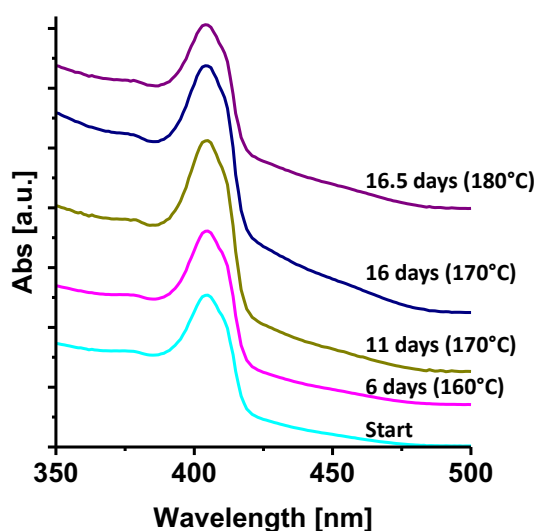
To confirm the critical role of Cd(OAc)<sub>2</sub> in the Ostwald ripening experiments, we conducted a control reaction with the same concentration of “baby” CdS NPLs but without Cd(OAc)<sub>2</sub>. The “baby” CdS NPLs gradually dissolved, and no thicker NPLs appeared (Figure S4).



**Figure S4.** Absorption spectra (left) for aliquots taken from an Ostwald ripening experiment conducted at 180 °C without added Cd(OAc)<sub>2</sub>. The absorption peaks gradually disappear as the ripening proceeds with time. TEM micrographs (right) show that nanocrystals with random sizes and shapes are formed.

### S4. Ripening Laterally Extended 4-ML CdS NPLs

We heated laterally extended 4-ML CdS NPLs under similar conditions as the “baby” CdS NPLs. As can be seen from the absorption spectra in Figure S5, no transitions to thicker populations were observed over the experimental time frame (16.5 days).

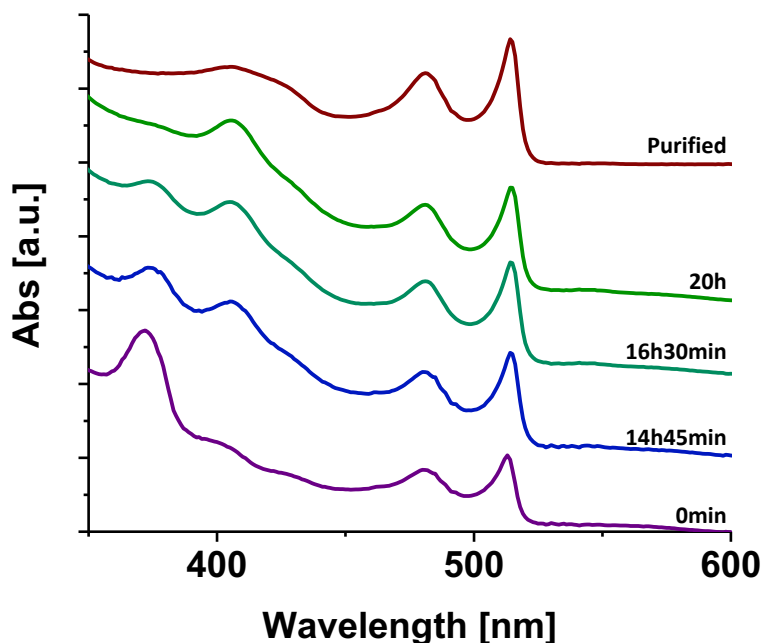


**Figure S5.** Absorption spectra of laterally extended 4-ML CdS NPLs heated in the presence of Cd(OAc)<sub>2</sub>·2H<sub>2</sub>O versus time. After 6 days, the temperature was raised to 170°C, and after 16 days to 180°C.

## S5. Synthesis of CdSe–CdS Core–Crown NPLs

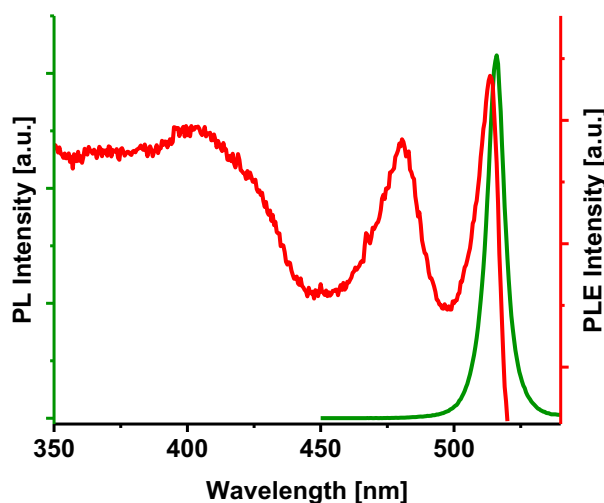
When 2- and 3-ML “baby” CdS NPLs were mixed with laterally extended 4-ML CdSe NPLs and heated to 160 °C, we observed a continuous dissolution of the “baby” CdS NPLs. This is seen in the absorption spectra shown in Figure S6. At the same time, absorption peaks for the 4-ML CdSe NPLs remained, and new peaks for 4-ML CdS NPLs appeared. After 20 h, the disappearance of the 2- and 3-ML “baby” CdS NPLs was complete.

Note that the absorption feature at 405 nm for the 4-ML CdS NPLs is significantly reduced after washing of the mixture with small amounts of ethanol. This is due to selective precipitation of the pure 4-ML CdS NPLs during the purification process.



**Figure S6.** Absorption spectra of aliquots taken during the synthesis of CdSe–CdS core–crown NPLs from laterally extended CdSe NPLs and CdS “baby” NPLs.

However, the photoluminescence excitation (PLE) spectrum (Figure S7), which was taken by monitoring the PL at the CdSe emission peak at 520 nm, shows that an absorption feature at 405 nm is still present after purification. This indicates that absorption due to 4-ML CdS NPLs is leading to emission from the 4-ML CdSe NPLs. This is consistent with CdSe–CdS core–crown NPLs. In other words, the CdS crown absorbs at 405 nm, the excitation is transferred to the CdSe core, and emission occurs at 520 nm.



**Figure S7.** PL (excitation at 400 nm) and PLE (emission at 520 nm) spectra of CdSe–CdS core–crown NPLs synthesized via ripening. See discussion above and that related to Figure 3 in the main text for further details.

## S6. Supplementary References

(1) Mahler, B.; Nadal, B.; Bouet, C.; Patriarche, G.; Dubertret, B., Core/Shell Colloidal Semiconductor Nanoplatelets. *J. Am. Chem. Soc.* **2012**, *134*, 18591–18598.